



## Effects of Bi<sup>3+</sup> doping on the optical properties of Er<sup>3+</sup>:Y<sub>2</sub>O<sub>3</sub>

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### ARTICLE INFO

#### Article history:

Received 22 June 2010

Received in revised form 9 September 2010

Accepted 18 September 2010

Available online 25 September 2010

#### Keywords:

Luminescence

Optical materials

Optical properties

Sol–gel process

### ABSTRACT

The influences of Bi<sup>3+</sup> doping on the optical properties of Er<sup>3+</sup>:Y<sub>2</sub>O<sub>3</sub> are investigated under UV and IR excitations. The emission intensity of Er<sup>3+</sup> is remarkably enhanced by the introduction of Bi<sup>3+</sup> under both two excitations. The emission enhancement under UV excitation originates from the energy transfer from Bi<sup>3+</sup> to Er<sup>3+</sup>, while under IR excitation it can be attributed to the modification of the local crystal field around the Er<sup>3+</sup>.

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### 1. Introduction

Over the past several decades, considerable works have been carried out on the rare earth ions doped luminescent materials regarding their applications in illumination, color displays, and biomedical imaging [1–3]. Among luminescent rare earth ions, Er<sup>3+</sup> is one of the most popular efficient ions, because its excited states (<sup>4</sup>I<sub>11/2</sub>) with long lifetimes can be easily populated by 980 nm irradiation [4,5]. As for the host for luminescent materials, most attention has been paid to oxide ceramics and nanopowders recently [6–8]. Y<sub>2</sub>O<sub>3</sub> is a promising oxide host material due to its high melting point, high chemical stability and low phonon energy. Er<sup>3+</sup>:Y<sub>2</sub>O<sub>3</sub> has been established as one of the model systems for generating efficient upconversion radiation under 980 nm laser excitation [4,5].

As the emission of most luminescent materials does not have enough intensity, it is still a formidable challenge about how to improve the emission efficiency of these materials. It is known that doping of sensitizer ions or breaking of the crystal field symmetry can lead to the increase in the emission intensity of rare earth ions [9–11]. Bi<sup>3+</sup> possesses a broad absorption band centering around 330 nm, and a broad emission band ranging from 300 to 700 nm

[12–14]. The emission band of Bi<sup>3+</sup> overlaps with the <sup>4</sup>I<sub>15/2</sub> → <sup>4</sup>F<sub>7/2</sub> and <sup>4</sup>I<sub>15/2</sub> → <sup>2</sup>H<sub>11/2</sub> transitions of Er<sup>3+</sup>, so Bi<sup>3+</sup> can also be used as a sensitizer of Er<sup>3+</sup> for the radiative energy transfer process. The broad absorb band of Bi<sup>3+</sup> is beneficial to absorb more energy and enhance Er<sup>3+</sup> emission under UV excitation. Moreover, as the radius of Bi<sup>3+</sup> is larger than that of Y<sup>3+</sup>, substitution of Y<sup>3+</sup> with Bi<sup>3+</sup> can modify the symmetry of crystal field in Y<sub>2</sub>O<sub>3</sub>, which is in favor of the enhancement of emission intensity.

In this study, Bi<sup>3+</sup> is selected as codopant to enhance the luminescent efficiency of Er<sup>3+</sup>. The emission intensity of Er<sup>3+</sup> is increased under both 330 nm and 980 nm excitations. The relevant mechanisms are discussed in detail.

### 2. Experimental

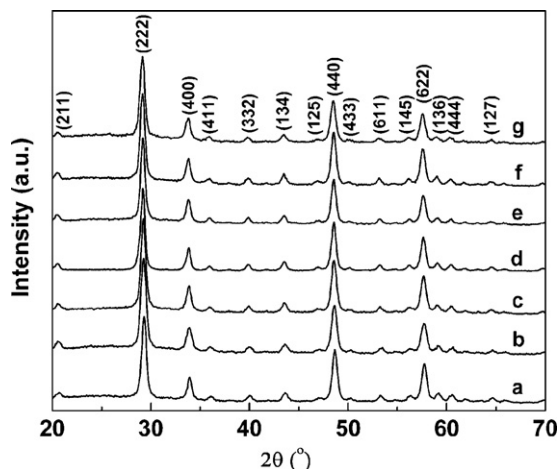
Y<sub>2</sub>O<sub>3</sub> powders doped with different concentrations of Er<sup>3+</sup> and Bi<sup>3+</sup> were prepared by a sol–gel combustion method. High purity Y<sub>2</sub>O<sub>3</sub>, Er<sub>2</sub>O<sub>3</sub>, Bi<sub>2</sub>O<sub>3</sub> and C<sub>6</sub>H<sub>8</sub>O<sub>7</sub>·H<sub>2</sub>O were used as starting materials. First, stoichiometric Y<sub>2</sub>O<sub>3</sub>, Er<sub>2</sub>O<sub>3</sub> and Bi<sub>2</sub>O<sub>3</sub> were dissolved in HNO<sub>3</sub>. Then C<sub>6</sub>H<sub>8</sub>O<sub>7</sub>·H<sub>2</sub>O was added to the mixed solution with the molar ratio of nitrate to citric acid to be 4:1. The resulting solution was heated at 80 °C in order to obtain a gel. Subsequently, the gel was rapidly heated to 200 °C and an autocombustion process took place. Finally, the precursor was calcined at 800 °C in air to obtain Er<sup>3+</sup> and Bi<sup>3+</sup> codoped Y<sub>2</sub>O<sub>3</sub> powders. The XRD patterns of the powders were recorded by a Rigaku D/max-γB diffractometer using Cu Kα radiation (λ = 0.15418 nm). The content of Bi<sup>3+</sup> was determined by EDS measurement, and the results show that the Bi<sup>3+</sup> is nearly in stoichiometric. The Stokes emission measurements were performed on Hitachi F-4500 Fluorescence Spectrophotometer at room temperature. The upconversion emission spectra were measured by a power controllable 980 diode laser and detected with a lens-coupled monochromator with an attached photomultiplier.

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**Table 1**  
The lattice parameter of  $Y_2O_3$  doped with different contents of  $Er^{3+}$  and  $Bi^{3+}$ : (a)  $Bi^{3+}:Y_2O_3$ ; (b)  $Er^{3+}:Y_2O_3$ ; (c) 1.0 mol%  $Bi^{3+}$  codoped  $Er^{3+}:Y_2O_3$ ; (d) 1.5 mol%  $Bi^{3+}$  codoped  $Er^{3+}:Y_2O_3$ ; (e) 2.0 mol%  $Bi^{3+}$  codoped  $Er^{3+}:Y_2O_3$ ; (f) 2.5 mol%  $Bi^{3+}$  codoped  $Er^{3+}:Y_2O_3$ ; (g) 3.0 mol%  $Bi^{3+}$  codoped  $Er^{3+}:Y_2O_3$ .

Samples	a	b	c	d	e	f	g
Lattice parameter (Å)	10.606	10.603	10.605	10.608	10.610	10.615	10.618

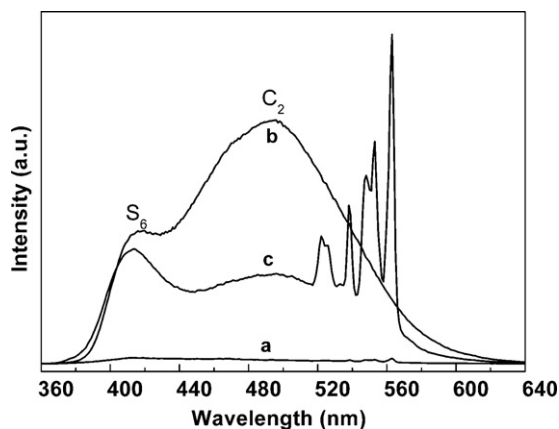


**Fig. 1.** XRD patterns of  $Y_2O_3$  powders doped with different concentrations of  $Er^{3+}$  and  $Bi^{3+}$ : (a)  $Bi^{3+}:Y_2O_3$ ; (b)  $Er^{3+}:Y_2O_3$ ; (c) 1.0 mol%  $Bi^{3+}$  codoped  $Er^{3+}:Y_2O_3$ ; (d) 1.5 mol%  $Bi^{3+}$  codoped  $Er^{3+}:Y_2O_3$ ; (e) 2.0 mol%  $Bi^{3+}$  codoped  $Er^{3+}:Y_2O_3$ ; (f) 2.5 mol%  $Bi^{3+}$  codoped  $Er^{3+}:Y_2O_3$ ; (g) 3.0 mol%  $Bi^{3+}$  codoped  $Er^{3+}:Y_2O_3$ .

### 3. Results and discussion

The XRD patterns of  $Y_2O_3$  doped with different concentrations of  $Er^{3+}$  and  $Bi^{3+}$  are shown in Fig. 1. The XRD data confirm that all samples are single phase with cubic structure (JPDFS No. 86-1107). The lattice parameters are listed in Table 1. The results show that  $Bi^{3+}$  and  $Er^{3+}$  were successfully doped into  $Y_2O_3$  host, and with the increase in  $Bi^{3+}$  content, the lattice parameter increases since the ion radius of  $Bi^{3+}$  is larger than that of  $Y^{3+}$ . It can also be found that the FWHM of the peaks in XRD patterns does not vary with the increase in  $Bi^{3+}$  content, namely,  $Bi^{3+}$  doping does not influence the crystallinity of  $Y_2O_3$  which could affect the emission of  $Er^{3+}$ .

Fig. 2 shows the emission spectra of 1.0 mol%  $Er^{3+}:Y_2O_3$  ( $Er^{3+}:Y_2O_3$ ), 1.0 mol%  $Bi^{3+}:Y_2O_3$  ( $Bi^{3+}:Y_2O_3$ ) and 1.0 mol%  $Bi^{3+}$  codoped  $Er^{3+}:Y_2O_3$ . The strong emission band of  $Bi^{3+}$  locates at wavelengths from 360 to 630 nm, while the emission peaks of  $Er^{3+}:Y_2O_3$  are very weak. However, the 1.0 mol%  $Bi^{3+}$  activated  $Er^{3+}:Y_2O_3$  exhibits sharp peaks due to  ${}^2H_{11/2}/{}^4S_{3/2} \rightarrow {}^4I_{15/2}$  transi-

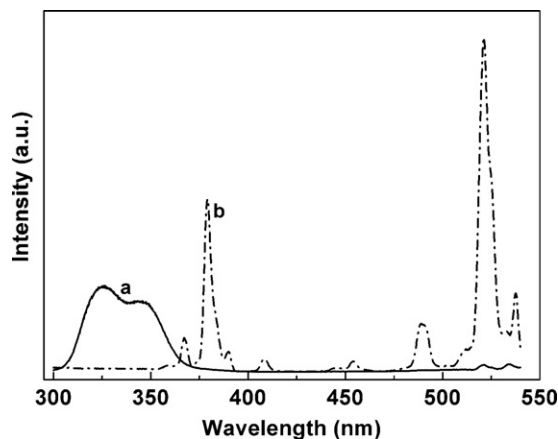


**Fig. 2.** Emission spectra of  $Er^{3+}:Y_2O_3$  (a, excited at 330 nm),  $Bi^{3+}:Y_2O_3$  (b, excited at 330 nm) and 1.0 mol%  $Bi^{3+}$  codoped  $Er^{3+}:Y_2O_3$  (c, excited at 330 nm).

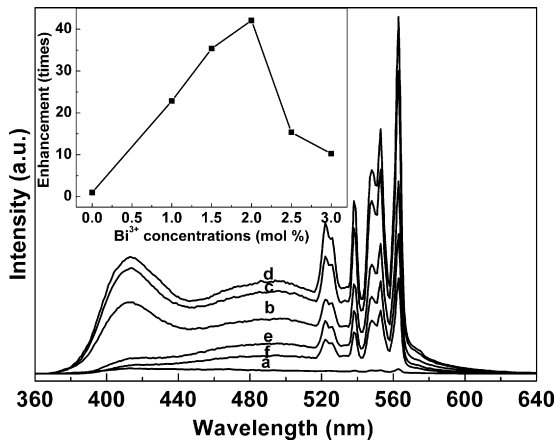
tions of  $Er^{3+}$  ions and its emission intensity is about 42 times larger than that of the sample without  $Bi^{3+}$ .

There are two main mechanisms for energy transfer between sensitizer and activator: (1) radiative transfer through emission of sensitizer and reabsorption by activator; (2) non-radiative transfer associated with resonance between absorber and emitter [13]. As shown in Fig. 3, the strong excitation band from 300 to 360 nm is attributed to the  ${}^1S_0 \rightarrow {}^3P_1$  transition of  $Bi^{3+}$  while the excitation peaks at wavelength longer than 360 nm are ascribed to the  ${}^4I_{15/2} \rightarrow {}^4F_{7/2}$  and  ${}^4I_{15/2} \rightarrow {}^2H_{11/2}$  transitions of  $Er^{3+}$  [15,16]. The emission band of  $Bi^{3+}$  ranging from 360 to 630 nm significantly overlaps with the excitation spectrum of  $Er^{3+}$  ranging from 360 to 540 nm. Therefore, it can be concluded that in this system  $Bi^{3+}$  is a sensitizer for the luminescence of  $Er^{3+}$  and  $Bi^{3+}$  can transfer energy to  $Er^{3+}$  through the radiative transfer process. The luminescent intensity of  $Er^{3+}$  is remarkably enhanced by the incorporation of  $Bi^{3+}$  under 330 nm light excitation, which suggested a very efficient energy transfer from  $Bi^{3+}$  to  $Er^{3+}$ .

Fig. 4 shows the emission spectra of  $Er^{3+}:Y_2O_3$  codoped with different amounts of  $Bi^{3+}$  (excited at 330 nm). In Fig. 4, the sharp emission peaks correspond to the  ${}^2H_{11/2}/{}^4S_{3/2} \rightarrow {}^4I_{15/2}$  transitions of  $Er^{3+}$ . The position of emission peaks dose not vary with the  $Bi^{3+}$  concentration, but the luminescence intensity changes a lot. The inset displays the enhancement magnitude of the emission intensity of  $Er^{3+}$ , ranging from 515 to 580 nm, as a function of  $Bi^{3+}$  concentration. As shown in the inset, the intensity of  $Er^{3+}$  considerably increases first, reaches a maximum at the 2.0 mol%  $Bi^{3+}$ , and falls thereafter. The decrease in the emission intensity is caused by the concentration quenching of  $Bi^{3+}$ . The total energy transfer efficiency depends not only on the probability of energy transfer from sensitizer to activator, but also on that between the sensitizers [15,17]. The energy transfer efficiency increases with the decrease of the distance between the sensitizer and activator (or sensitizer). In this study,  $Bi^{3+}$  is the sensitizer and  $Er^{3+}$  is the activator. In the case of the  $Bi^{3+}$  content less than 2 mol%, the distance between  $Bi^{3+}$  and  $Er^{3+}$  decreases with  $Bi^{3+}$  doping, and thus the energy transfer efficiency from  $Bi^{3+}$  to  $Er^{3+}$  increases. However, when the concentration of  $Bi^{3+}$  exceeds 2.0 mol%, the average distance between two  $Bi^{3+}$  becomes short. This leads to the efficient energy transfer



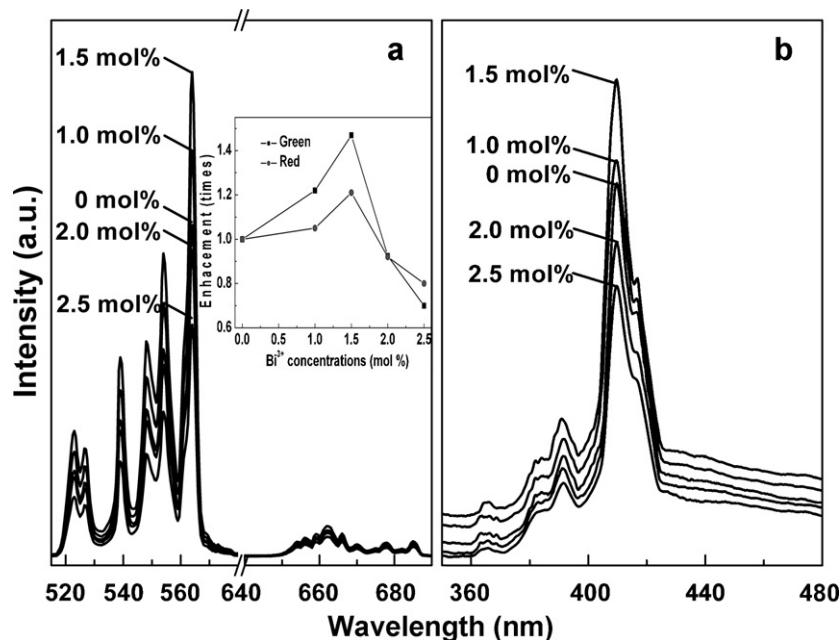
**Fig. 3.** Excitation spectra of  $Bi^{3+}:Y_2O_3$  (a, monitored at 563 nm),  $Er^{3+}:Y_2O_3$  (b, monitored at 563 nm).



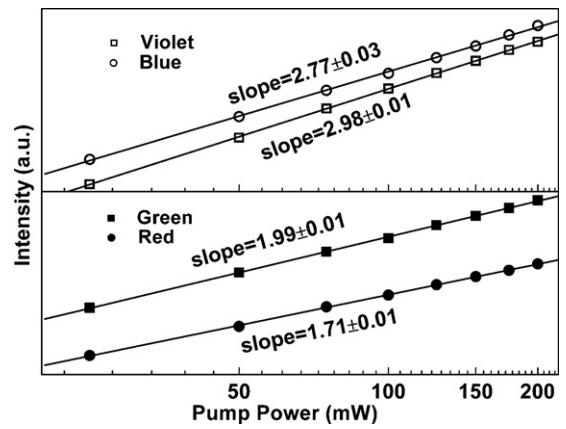
**Fig. 4.** Emission spectra of the  $\text{Bi}^{3+}$  and  $\text{Er}^{3+}$  codoped  $\text{Y}_2\text{O}_3$  phosphors (with a fixed  $\text{Er}^{3+}$  concentration of 1.0 mol%, excited at 330 nm): (a) 0 mol%  $\text{Bi}^{3+}$ ; (b) 1.0 mol%  $\text{Bi}^{3+}$ ; (c) 1.5 mol%  $\text{Bi}^{3+}$ ; (d) 2.0 mol%  $\text{Bi}^{3+}$ ; (e) 2.5 mol%  $\text{Bi}^{3+}$ ; (f) 3.0 mol%  $\text{Bi}^{3+}$ , the inset is the emission intensity of  $\text{Er}^{3+}$  as a function of the  $\text{Bi}^{3+}$  concentration.

between  $\text{Bi}^{3+}$ , and the non-radiative process increases gradually, which hinders the energy transfer from  $\text{Bi}^{3+}$  to  $\text{Er}^{3+}$ . In a word, adding an optimal amount of  $\text{Bi}^{3+}$  to  $\text{Er}^{3+}:\text{Y}_2\text{O}_3$  can enhance the emission intensity of  $\text{Er}^{3+}$  through energy transfer from  $\text{Bi}^{3+}$  to  $\text{Er}^{3+}$ , but after exceeding the optimum concentration, the energy transfer between  $\text{Bi}^{3+}$  becomes more efficient which thus results in less energy transfer from  $\text{Bi}^{3+}$  to  $\text{Er}^{3+}$ .

As discussed above, an appropriate amount of  $\text{Bi}^{3+}$  doping can enhance emission intensity of  $\text{Er}^{3+}:\text{Y}_2\text{O}_3$  through the radiative transfer process under UV excitation. In addition to UV excitation, the emission intensity of  $\text{Er}^{3+}:\text{Y}_2\text{O}_3$  under IR excitation can also be increased by  $\text{Bi}^{3+}$  doping. Fig. 5a displays the upconversion luminescence spectra of the  $\text{Er}^{3+}/\text{Bi}^{3+}$  codoped  $\text{Y}_2\text{O}_3$  under 980 nm excitation. The inset shows the integral intensity of green and red emissions as a function of  $\text{Bi}^{3+}$  content. As shown in the inset, both the green and red emissions increase as the  $\text{Bi}^{3+}$  content is below 1.5 mol%, however, above that concentration the emission intensity becomes weak. In order to clarify the mechanism of the change



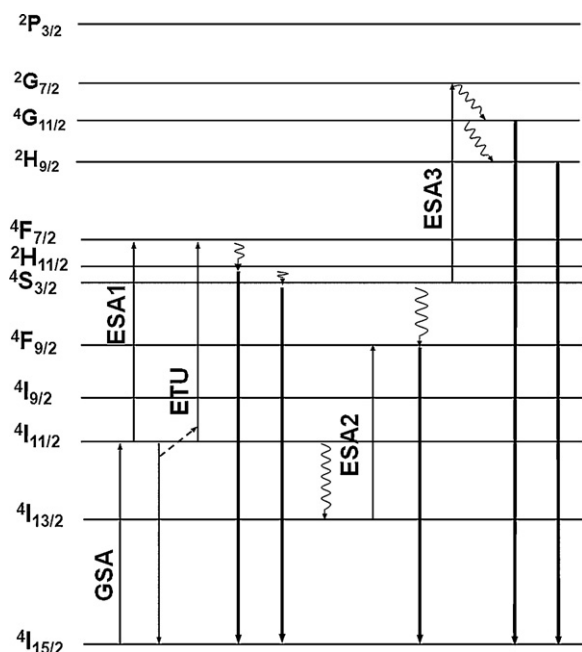
**Fig. 5.** Upconversion emissions of  $\text{Er}^{3+}:\text{Y}_2\text{O}_3$  powders with different concentrations of  $\text{Bi}^{3+}$  under 980 nm excitation. The inset is the emission intensity as a function of the  $\text{Bi}^{3+}$  concentration.



**Fig. 6.** The pump power dependence of the violet, blue, green and red upconversion emission intensity of 1.5 mol%  $\text{Bi}^{3+}$  codoped  $\text{Er}^{3+}:\text{Y}_2\text{O}_3$ .

in green and red emissions depending on  $\text{Bi}^{3+}$  content, we measure the violet and blue emissions of  $\text{Er}^{3+}/\text{Bi}^{3+}$  codoped  $\text{Y}_2\text{O}_3$  under excitation at 980 nm as well as the power dependence of violet, blue, green and red emissions. It is known that this dependence is expressed as  $I \propto nP$ , where  $I$  is the intensity of emission,  $P$  is the laser pump power, and  $n$  is the number of phonons needed to produce the emission [18].

Fig. 5b shows the violet and blue spectra of  $\text{Er}^{3+}/\text{Bi}^{3+}$  codoped  $\text{Y}_2\text{O}_3$  under excitation at 980 nm. We cannot observe the emission band of  $\text{Bi}^{3+}$  in the upconversion spectra ranging from 350 to 480 nm. Moreover, as shown in Fig. 6, the values of slopes for the violet, blue, green and red emissions are 2.98, 2.77, 1.99 and 1.71, respectively. It indicates that both the violet and blue emissions are 3 phonon processes, and both the green and red emissions are 2 phonon processes. Consequently, the upconversion processes of  $\text{Er}^{3+}/\text{Bi}^{3+}$  codoped  $\text{Y}_2\text{O}_3$  could be described as Fig. 7. The  $\text{Er}^{3+}$  can be promoted to the  $^4\text{I}_{11/2}$  state through ground state absorption (GSA) process, and then nonradiative relaxation occurs and populates the  $^4\text{I}_{13/2}$  state. Subsequently,  $^4\text{I}_{11/2}$  and  $^4\text{I}_{13/2}$  state are further excited to the  $^4\text{F}_{7/2}$  and  $^4\text{F}_{9/2}$  state respectively via excited state



**Fig. 7.** Energy level diagram of  $\text{Er}^{3+}$  ion as well as the upconversion processes of the violet, blue, green and red emissions in 1.5 mol%  $\text{Bi}^{3+}$  codoped  $\text{Er}^{3+}:\text{Y}_2\text{O}_3$  under 980 nm excitation.

absorption (ESA1 and ESA2) or energy transfer upconversion (ETU) processes. After that nonradiative relaxation processes populate the  $^2\text{H}_{11/2}/^4\text{S}_{3/2}$  and  $^4\text{F}_{9/2}$  states, so the green and red emissions are observed by the transitions of  $^2\text{H}_{11/2}/^4\text{S}_{3/2} \rightarrow ^4\text{I}_{15/2}$  and  $^4\text{F}_{9/2} \rightarrow ^4\text{I}_{15/2}$ . The  $\text{Er}^{3+}$  ions in the  $^4\text{S}_{3/2}$  state are excited to  $^2\text{G}_{7/2}$  state by absorbing a third phonon (ESA3), and then nonradiatively relax to  $^4\text{G}_{11/2}$ ,  $^2\text{H}_{9/2}$  states. From these two states violet and blue emissions are emitted. As mentioned above, the doping of  $\text{Bi}^{3+}$  does not change the emission process of  $\text{Er}^{3+}$ , that is to say, there is not energy transfer between  $\text{Bi}^{3+}$  and  $\text{Er}^{3+}$  under 980 nm excitation. Moreover, high crystallinity could lessen nonradiative relaxation probability and then enhance the emission intensity of  $\text{Er}^{3+}$ . Nevertheless, from the XRD patterns we can conclude that  $\text{Bi}^{3+}$  doping does not influence the crystallinity of  $\text{Er}^{3+}/\text{Bi}^{3+}$  codoped  $\text{Y}_2\text{O}_3$ . Therefore, the enhanced emission of  $\text{Er}^{3+}/\text{Bi}^{3+}$  codoped  $\text{Y}_2\text{O}_3$  originates from neither the energy transfer nor the crystallinity. It is well known that the intra- $4f$  electronic transitions of rare earth are parity forbidden according to the quantum mechanical selection rules. However, the forbiddance can be partially broken when the rare earth situates at low symmetry sites [10,19]. The radius of  $\text{Bi}^{3+}$  (1.03 Å) is larger than that of  $\text{Y}^{3+}$  (0.90 Å) [20], so the substitution of  $\text{Bi}^{3+}$  for  $\text{Y}^{3+}$  can modify the symmetry of the crystal field in the lattice and enhance the radiative transition rate favoring the enhancement of upconversion emission intensity. The sensitization of upconversion emission of  $\text{Er}^{3+}$  on  $\text{Bi}^{3+}$  doping suggests that the  $\text{Bi}^{3+}$  could effectively modify the crystal field around  $\text{Er}^{3+}$  and the modification of crystal field depends strongly on the  $\text{Bi}^{3+}$  doping content. The optimal doping content of  $\text{Bi}^{3+}$  is found to be 1.5 mol%, and the corresponding emission intensity is 1.5 times larger than that without  $\text{Bi}^{3+}$  doping.

When the  $\text{Bi}^{3+}$  content exceeds 1.5 mol%, the upconversion emission becomes weak due to the concentration quenching. For higher content of  $\text{Bi}^{3+}$ ,  $\text{Bi}_n^{3+}$  aggregates may be formed [21], which act as trapping centers and dissipate the absorbed energy nonradiatively instead of transferring it to the  $\text{Er}^{3+}$  activator.

#### 4. Conclusions

The enhancement of the emission in  $\text{Er}^{3+}/\text{Bi}^{3+}$  codoped  $\text{Y}_2\text{O}_3$  under 330 nm and 980 nm results from two different mechanisms. The remarkably enhanced luminescent intensity under UV excitation originates from the efficient energy transfer from  $\text{Bi}^{3+}$  to  $\text{Er}^{3+}$ , whereas the enhancement in the upconversion emissions under IR excitation is attributed to the modification of the local crystal field around the  $\text{Er}^{3+}$ . The significant enhancement of emission intensity under both UV and IR excitations suggests that  $\text{Er}^{3+}/\text{Bi}^{3+}:\text{Y}_2\text{O}_3$  could be a promising candidate in the application of optoelectronic devices.

#### Acknowledgments

This work was supported by the National Natural Science Foundation of China (Grant Nos. 50672019 and 10804024) and the Scientific Research Foundation for the Returned Overseas Chinese Scholars, State Education Ministry. Supported by “the Fundamental Research Funds for the Central Universities” (Grant No. HIT.NSRIF.2009056).

#### References

- [1] X.D. Qi, C.M. Liu, C.C. Kuo, *J. Alloys Compd.* 492 (2010) L61.
- [2] B.J. Chen, E.Y.B. Pun, H. Lin, *J. Alloys Compd.* 479 (2009) 352.
- [3] S.F. Lim, R. Riehn, W.S. Ryu, N. Khanarian, C.K. Tung, D. Tank, R.H. Austin, *Nano Lett.* 6 (2006) 169.
- [4] Q. Dong, G.J. Zhao, D.H. Cao, J.Y. Chen, Y.C. Ding, *J. Alloys Compd.* 493 (2010) 661.
- [5] F. Vetrone, J.C. Boyer, J.A. Capobianco, A. Speghini, M. Bettinelli, *Chem. Mater.* 15 (2003) 2737.
- [6] R. Srinivasan, N.R. Yogamalar, J. Elanchezhian, R.J. Joseyphus, A.C. Bose, *J. Alloys Compd.* 496 (2010) 472.
- [7] Y.P. Li, J.H. Zhang, X. Zhang, Y.S. Luo, S.Z. Lu, Z.D. Hao, X.J. Wang, *J. Phys. Chem. C* 113 (2009) 17705.
- [8] H.X. Zhou, Q.H. Yang, J. Xu, H.J. Zhang, *J. Alloys Compd.* 471 (2009) 474.
- [9] Y.F. Bai, Y.X. Wang, G.Y. Peng, K. Yang, X.R. Zhang, Y.L. Song, *J. Alloys Compd.* 478 (2009) 676.
- [10] M.Z. Yang, Y. Sui, S.P. Wang, X.J. Wang, Y.Q. Sheng, Z.G. Zhang, T.Q. Lü, W.F. Liu, *Chem. Phys. Lett.* 492 (2010) 40.
- [11] T.S. Chan, C.C. Kang, R.S. Liu, L. Chen, X.N. Liu, J.J. Ding, J. Bao, C. Gao, *J. Comb. Chem.* 9 (2007) 343.
- [12] S. Neeraj, N. Kijima, A.K. Cheetham, *Solid State Commun.* 131 (2004) 65.
- [13] M.Q. Wang, X.P. Fan, G.H. Xiong, *J. Phys. Chem. Solids* 56 (1995) 859.
- [14] L.G. Jacobssohn, M.W. Blair, S.C. Tornga, L.O. Brown, B.L. Bennett, R.E. Muenchausen, *J. Appl. Phys.* 104 (2008) 124303.
- [15] Q.H. Xu, B.C. Lin, Y.L. Mao, *J. Lumin.* 128 (2008) 1965.
- [16] L. Chen, H.W. Zheng, J.G. Cheng, P. Song, G.T. Yang, G.B. Zhang, C. Wu, *J. Lumin.* 128 (2008) 2027.
- [17] K. Fujioka, T. Saiki, S. Motokoshi, Y. Fujimoto, H. Fujita, M. Nakatsuka, *J. Lumin.* 130 (2010) 455.
- [18] G.Y. Chen, Y. Liu, Y.G. Zhang, G. Somesfalean, Z.G. Zhang, Q. Sun, F.P. Wang, *Appl. Phys. Lett.* 91 (2007) 133103.
- [19] S.H. Shin, J.H. Kang, D.Y. Jeon, D.S. Zhang, *J. Lumin.* 114 (2005) 275.
- [20] R.D. Shannon, *Acta. Cryst.* A32 (1976) 751.
- [21] X.L. Pang, Y. Zhang, L.H. Ding, Z.H. Su, W.F. Zhang, *J. Nanosci. Nanotechnol.* 10 (2010) 1860.